

WDS 2015

Mössbauer Spectrometry of Magnetic Iron-Oxide–Based Nanoparticles

Denisa Kubániová



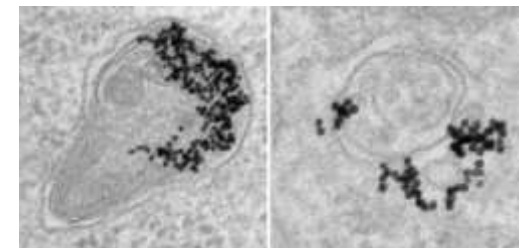
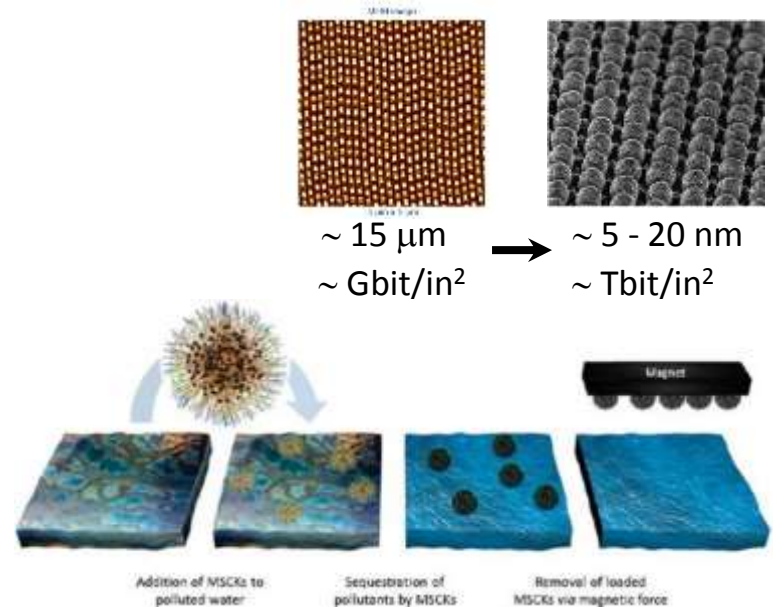
3rd of June 2015 Prague

Outline

- 1) Motivation for the study
- 2) Search for suitable materials
- 3) Mössbauer spectrometry as a tool to investigate cationic distribution over the crystallographic sites
- 4) Preparation of ϵ -phase of ferric oxide Fe_2O_3
- 5) Substituted ϵ - $\text{M}_x\text{Fe}_{2-x}\text{O}_3$
- 6) Substituted spinels $\text{M}_x\text{Fe}_{3-x}\text{O}_4$
- 7) Conclusions

Perspective applications of iron-oxide-based nanoparticles

- Electronics:
 - high-density magnetic data storage
 - mm-wave absorbers for high-speed WiFi
 - production of hydrogen gas
 - photovoltaics
 - gas sensors
- Catalysis
- Cleaning of polluted water (ferrofluids)
- Nanomedicine: *targetting, diagnosis, therapy*
 - contrast increase agents for MRI diagnostic (in vivo)
 - hyperthermia of tumors (in vivo)
 - targeted drug delivery, tracking and release (in vivo)
 - cell labeling and separation (in vitro)
 - every application request a material of specific properties

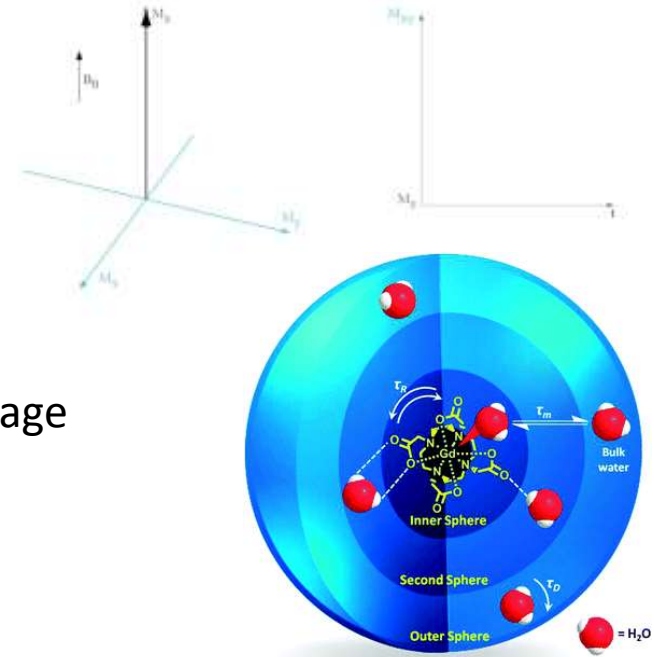


Nanomedical applications – MRI

Basic concept of MRI:

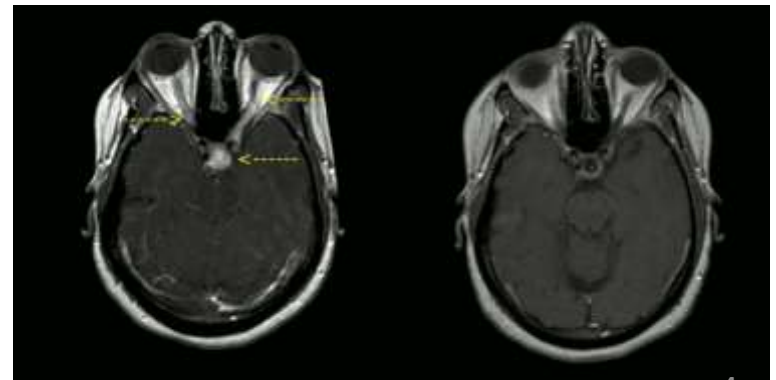
- T_1 spin-lattice relaxation $M_z = M_0 \cdot e^{-\frac{t}{T_1}}$
- T_2 spin-spin relaxation $M_{xy} = M_{xy0} \cdot e^{-\frac{t}{T_2}}$
- τ (pulse of B_{rf}) $\ll T_1, T_2 \rightarrow$ NMR signal
- pulse sequences $\rightarrow T_1, T_2$ of ^1H of water present in tissues
- various tissues differs in $T_1, T_2 \rightarrow$ contrast in MR image
- contrast agents (CA) affect T_1, T_2 by magnetic dipol interaction of ^1H spins with magnetization of CA

$$r_{2,MAR} = \frac{R_{2,MAR}}{[\text{Fe}]} = \frac{av_{mat}r^2\mu_0^2\gamma^2M^2}{9D}$$

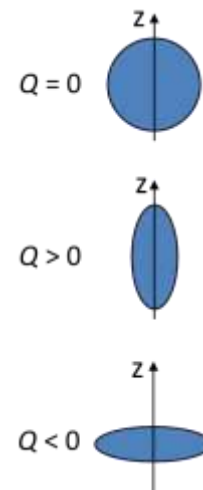
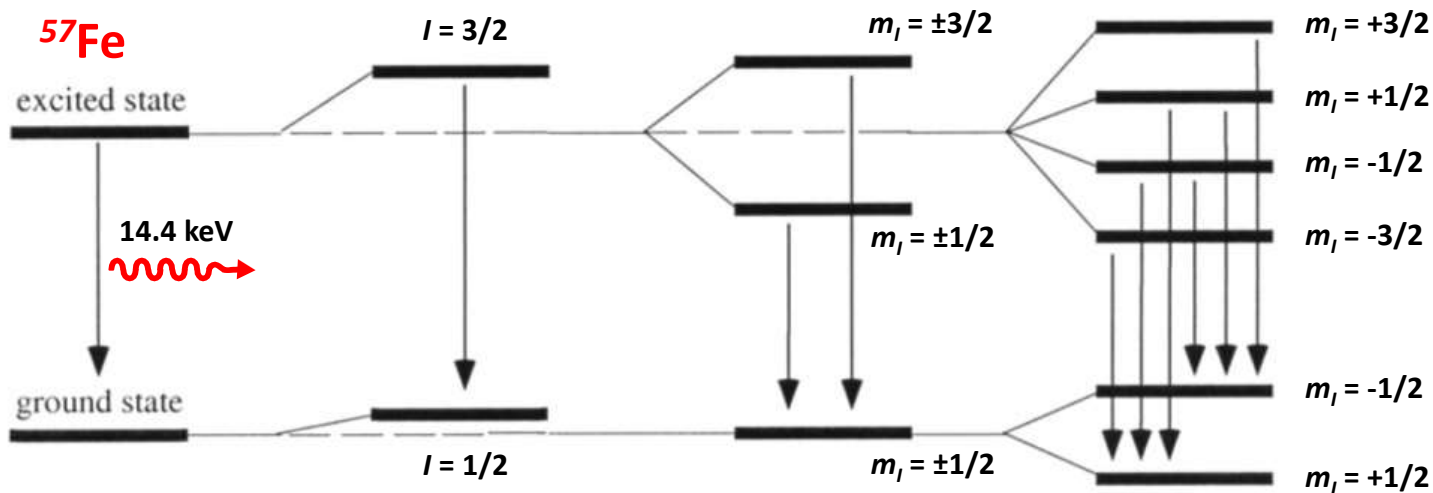


General requirements for nanoparticles:

- particle size (20-60 nm)
- narrow distribution
- reduced tendency to clustering
- stability in aqueous solution
- low toxicity, optimized biodegradability



Mössbauer spectroscopy (MS)



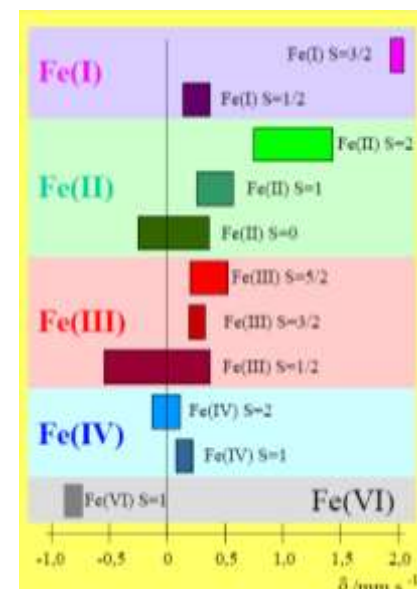
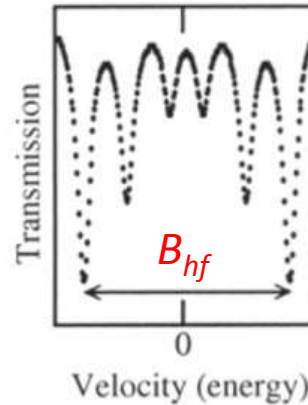
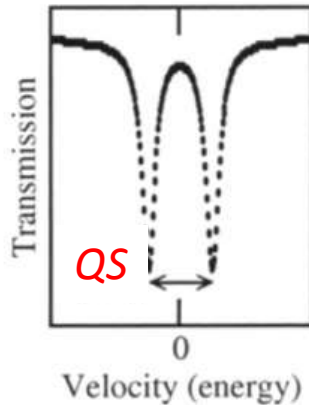
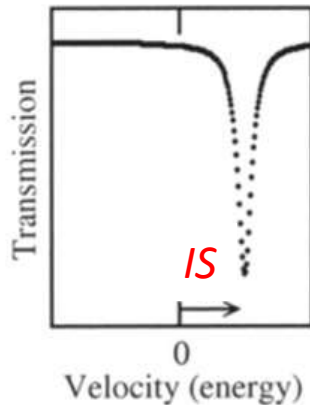
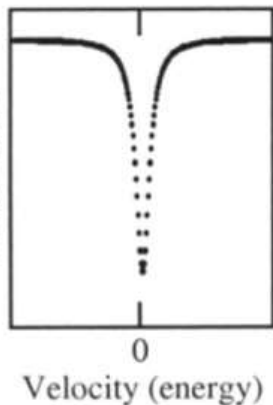
MS parameters:

unperturbed

isomer shift

quadrupole splitting

magnetic dipole



Nanoparticles of ϵ -phase Fe_2O_3

Methods of preparation:

- hydrothermal method
- microemulsion method
- sol-gel synthesis followed by heat treatment (HT)
- impregnation of mesoporous amorphous silica matrix (pores 5-7 nm) by nitrates $\text{Fe}(\text{NO}_3)_3$ + HT

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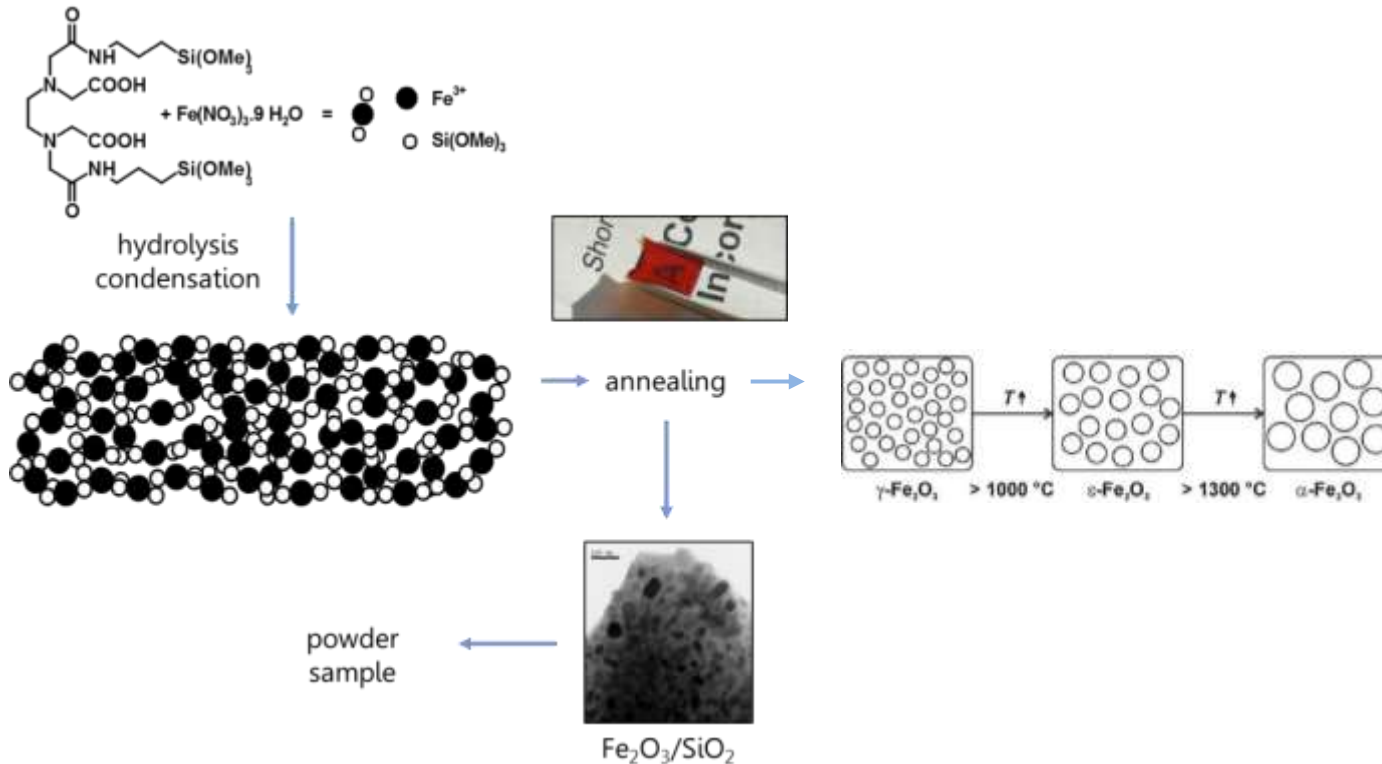
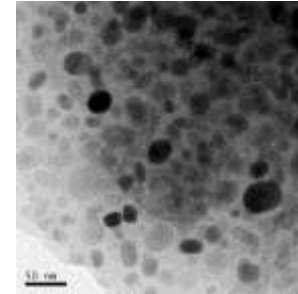
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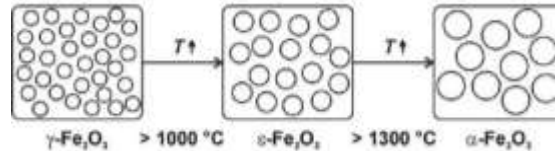
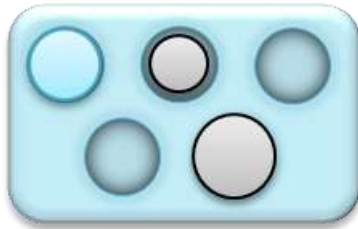
sol-gel



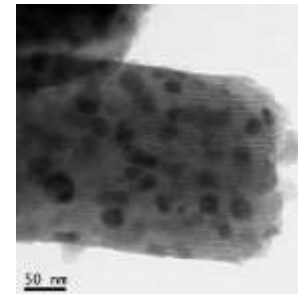
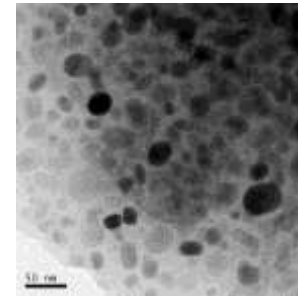
Nanoparticles of ϵ -phase Fe_2O_3

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sol-gel

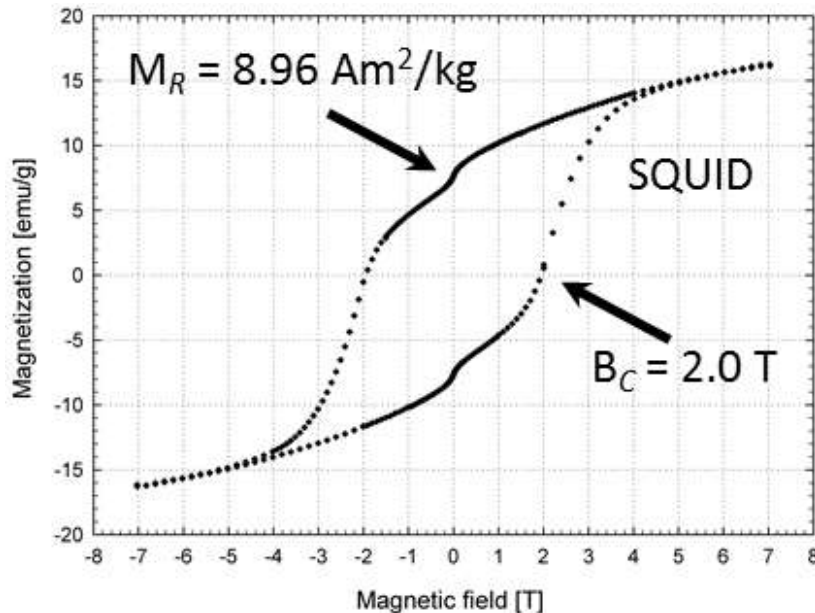


impregnation

Nanoparticles of ϵ -phase Fe_2O_3

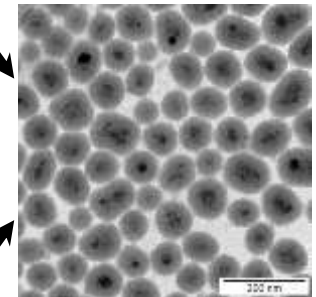
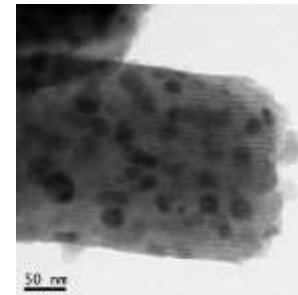
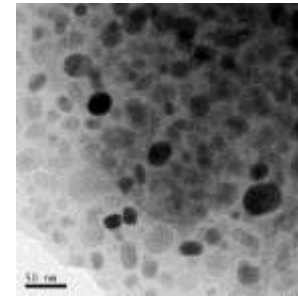
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Increase of magnetization by substitutions of Fe by diamagnetic ions.

sol-gel

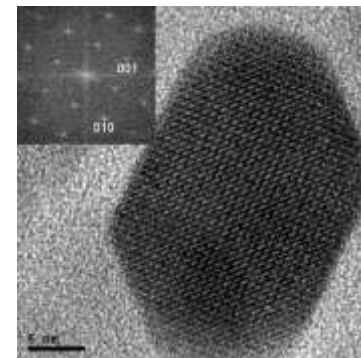


ϵ - Fe_2O_3 coated in SiO_2

Separation of NP:
 Optimal thickness?
 Restriction of agglomeration?
 Effect on relaxivity?

impregnation

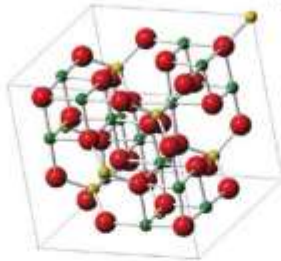
HRTEM of ϵ - Fe_2O_3 nanoparticle



MS of ϵ -Fe₂O₃ prepared by sol-gel method

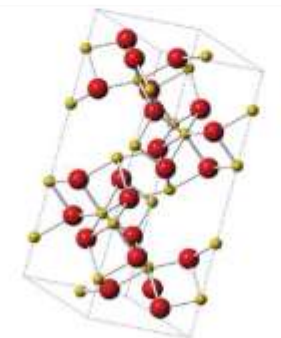
inverse spinel with cubic unit cell
a = 8.351 Å

(c) γ -Fe₂O₃



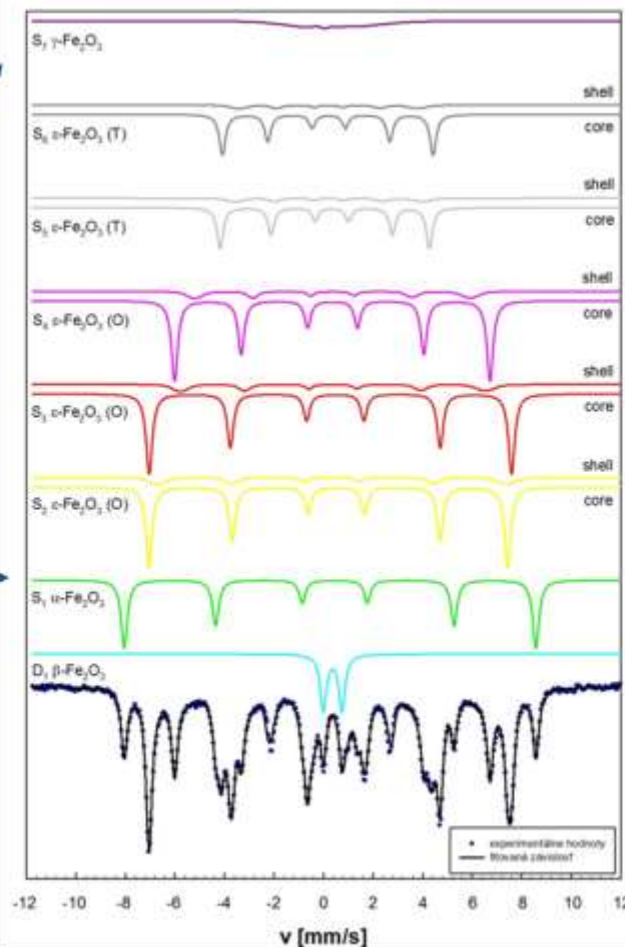
● Fe_A site
● Fe_B site
● O

(a) α -Fe₂O₃

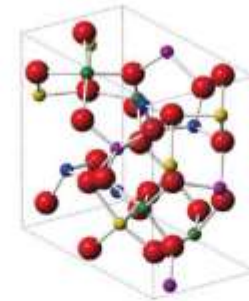


● Fe
● O

rhombohedrally centered hexagonal structure
T_N ~ 950 K
a = 5.0356 Å
c = 13.7489 Å



(d) ϵ -Fe₂O₃



orthorhombic structure

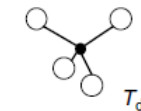
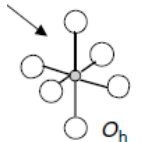
a = 5.095 Å

b = 8.789 Å

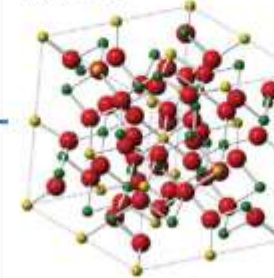
c = 9.437 Å

T_N ~ 490 K

● Fe₁ site
● Fe₂ site
● Fe₃ site
● Fe₄ site
● O



(b) β -Fe₂O₃



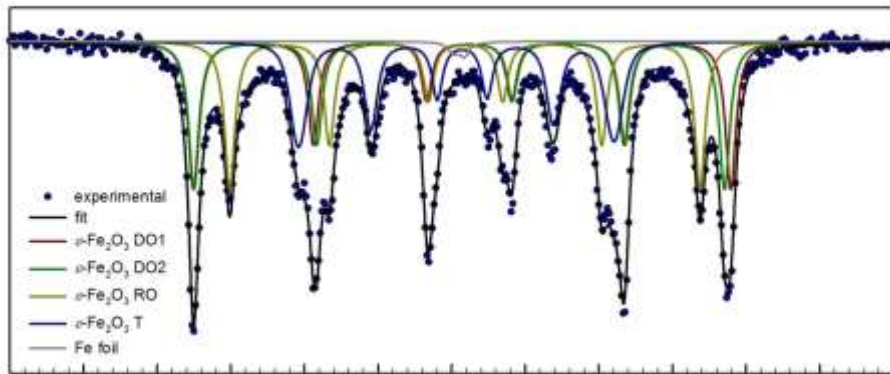
● b-site Fe
● d-site Fe
● O

body-centered cubic structure
a = 9.404 Å
T_N ~ 108 K

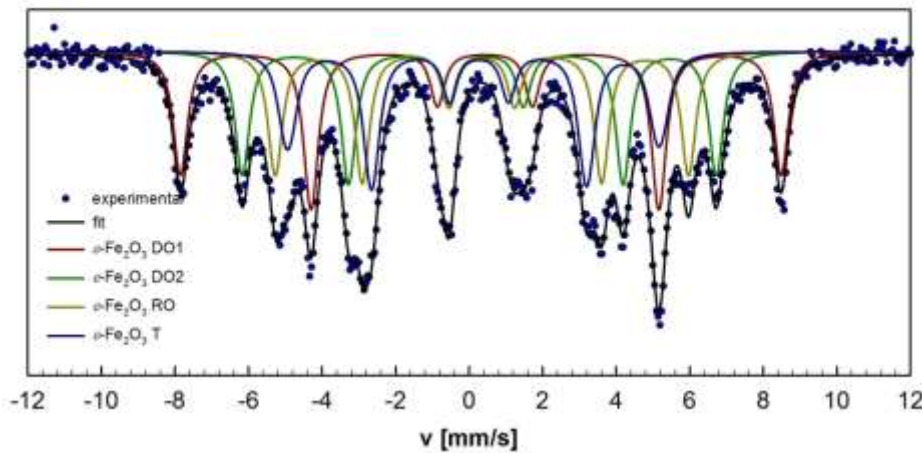
- using amorphous silica matrix was obtained max 92 % ϵ -Fe₂O₃
- restraint of spacial growth, prevention of agglomeration and formation of hematite
- other factors - iron concentration and content of organics in precursor, annealing temperature

MS of pure ϵ -Fe₂O₃ prepared by impregnation

ϵ -Fe₂O₃ nanoparticles (T=300 K, B=0 T)

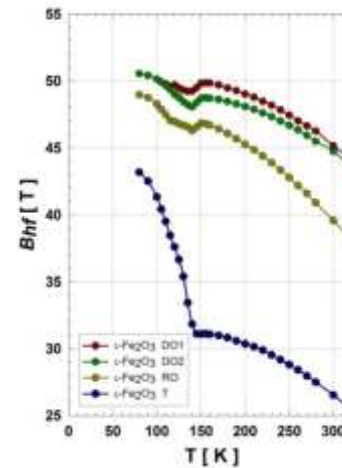


ϵ -Fe₂O₃ nanoparticles (T=300 K, B=6 T)

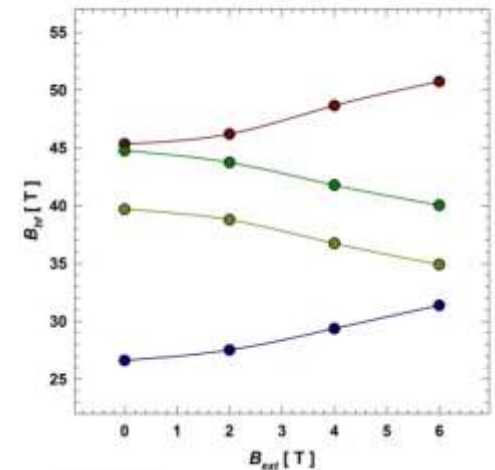


- separation of spectral components in Mössbauer spectra under the influence of an external field

Temperature dependence of B_{hf}



Magnetic field dependence of B_{hf} at T = 300 K



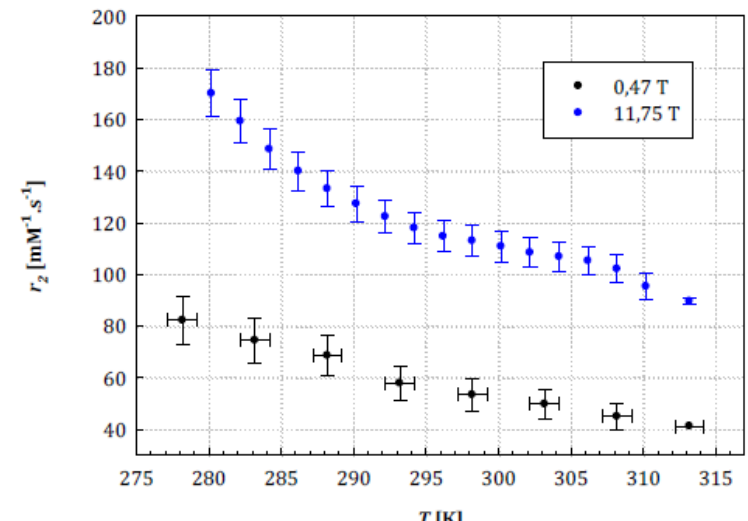
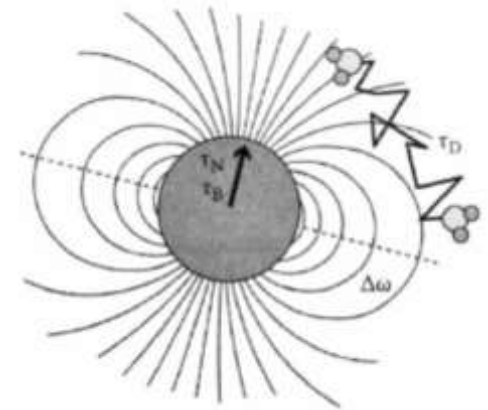
- using silica matrix with tube-like pores was obtained **99 % ϵ -Fe₂O₃**
- restraint of spacial growth to one direction crucial for formation of ϵ -Fe₂O₃, no β -Fe₂O₃ and minimal content of α -Fe₂O₃

MS of pure ϵ -Fe₂O₃ prepared by impregnation

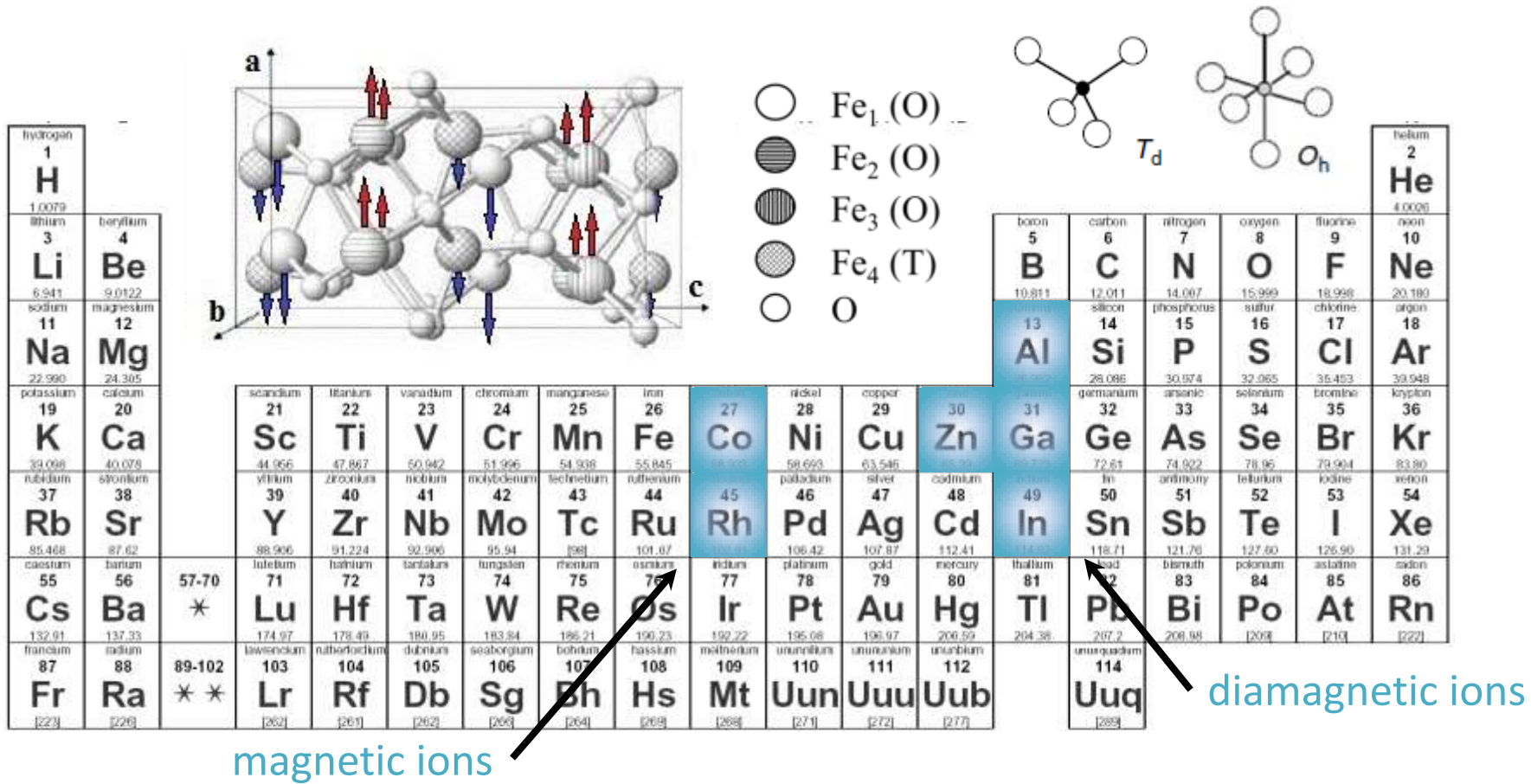
- ferrimagnetic nanoparticle in blocked state... static M
- Motional Averaging Regime (MAR), also outer-sphere regime

$$r_{2,MAR} = \frac{R_{2,MAR}}{[Fe]} = \frac{av_{mat}r^2\mu_0^2\gamma^2M^2}{9D}$$

- coated ϵ -Fe₂O₃ for T₂-weighted MRI
- $r_2(\epsilon$ -Fe₂O₃) – 2 to 5 times smaller than r_2 of commercial superparamagnetic γ -Fe₂O₃ or Fe₃O₄ due to lower magnetization in external fields
- can be improved by substitutions



Substituted $\epsilon\text{-M}_x\text{Fe}_{2-x}\text{O}_3$ for MRI



*Lanthanide series

| | | | | | | | | | | | | | |
|--|--------------------------------------|---|--|--|---------------------------------------|---------------------------------------|---|---------------------------------------|---|---|--------------------------------------|---|--|
| lanthanum 57 La 138.91 | cerium 58 Ce 140.12 | praseodymium 59 Pr 140.91 | neodymium 60 Nd 144.24 | promethium 61 Pm [145] | samarium 62 Sm 150.36 | europium 63 Eu 151.96 | gadolinium 64 Gd 157.25 | terbium 65 Tb 158.93 | dysprosium 66 Dy 162.50 | holmium 67 Ho 164.93 | erbium 68 Er 167.26 | thulium 69 Tm 168.93 | ytterbium 70 Yb 173.04 |
| actinium 89 Ac [227] | thorium 90 Th 232.04 | protactinium 91 Pa 231.04 | uranium 92 U 238.03 | neptunium 93 Np [237] | plutonium 94 Pu [244] | americium 95 Am [243] | curium 96 Cm [247] | berkelium 97 Bk [247] | californium 98 Cf [251] | einsteinium 99 Es [252] | fermium 100 Fm [257] | mendeleevium 101 Md [258] | nobelium 102 No [259] |

** Actinide series

Substituted ϵ - $M_x\text{Fe}_{2-x}\text{O}_3$ for MRI

- Ga, Al substitutions \rightarrow magnetization increased 2 to 4 times, modification of T_c in the range from 0 to 495 K
- Ga, Al, In substitutions \rightarrow increase of H_c
- distribution of ions over the crystallographic positions differs from that for bulk samples
- studies only for nanoparticles prepared by sol-gel method (ϵ -phase stabilized by Ba^{2+} and Ca^{2+} ions)

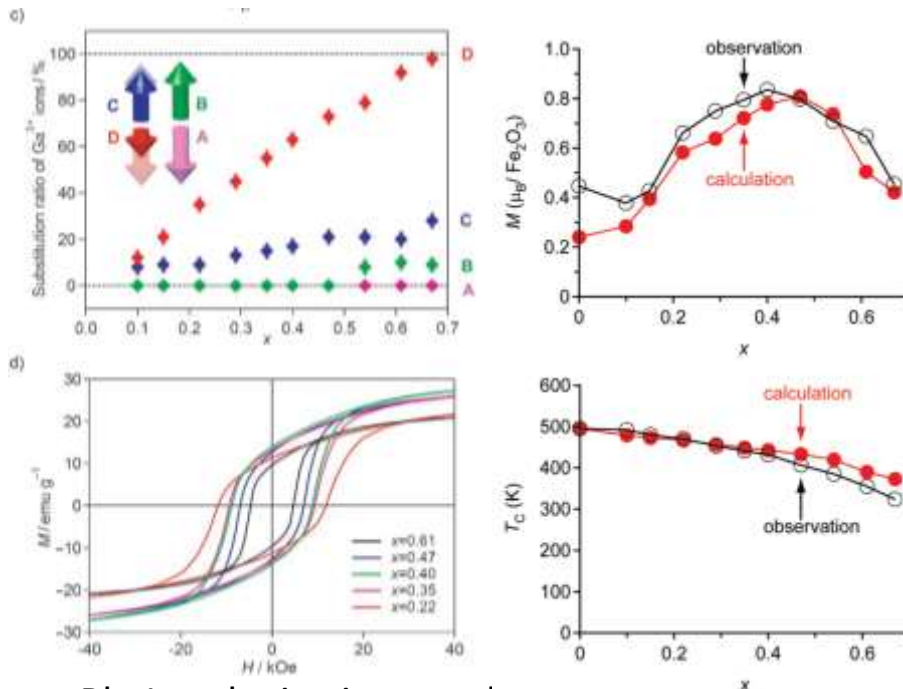
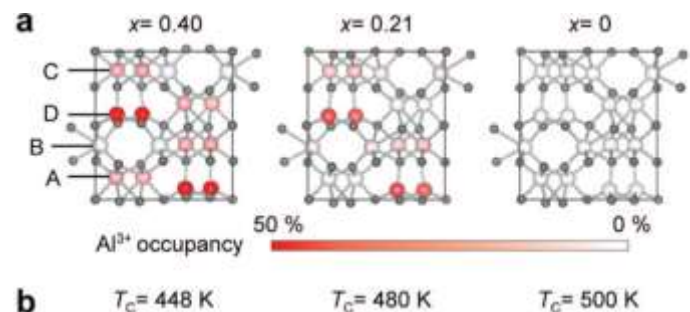


Table 2. Magnetic Properties of ϵ - $\text{Al}_x\text{Fe}_{2-x}\text{O}_3$

| x | T_c/K | H_c/kOe^a | $M_s/\text{emu g}^{-1a}$ |
|------|----------------|--------------------|--------------------------|
| 0 | 500 | 22.5 | 14.9 |
| 0.06 | 496 | 19.1 | 15.1 |
| 0.09 | 490 | 17.5 | 14.6 |
| 0.21 | 480 | 14.9 | 17.0 |
| 0.30 | 466 | 13.8 | 20.3 |
| 0.40 | 448 | 10.2 | 19.7 |

^a Values measured at 300 K.



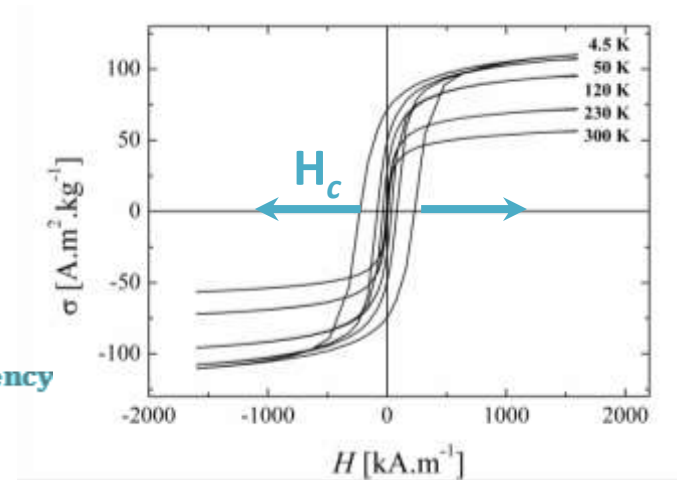
- Rh, In substitutions \rightarrow destabilization of ϵ - Fe_2O_3 structure, transformation to corundum structure

Nanomedical applications - Hyperthermia

- complementary cancer treatment
- local heating-up of tumor cells up to 41-46 °C
- SAR \approx hysteresis losses during repeated cycles (100-400 kHz)

$$\text{SAR} = 4,1868 \frac{P}{m_e} = C_e \frac{\Delta T}{\Delta t}$$

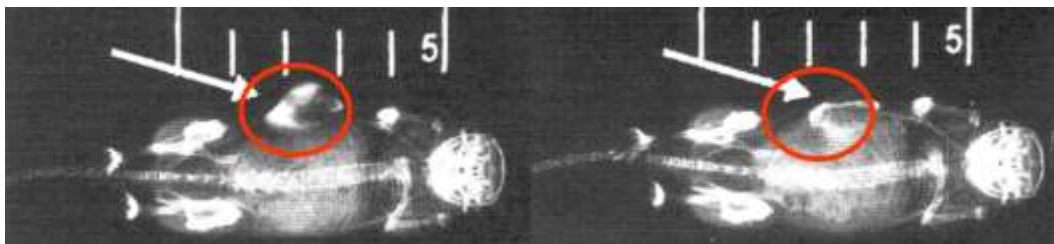
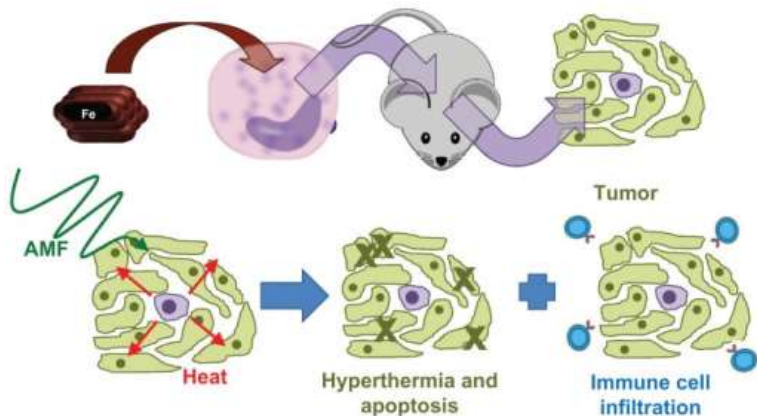
Power Dissipated = Area of B – H Curve \times Volume of Material \times Frequency



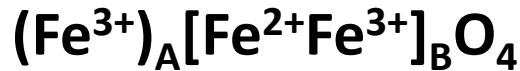
DC hysteresis loop of $\text{Co}_{0.5}\text{Zn}_{0.5}\text{Fe}_2\text{O}_4$ nanoparticles

Specific requirements:

- enhanced SAR – increase of H_c
- autoregulation by T_c adjustment (40-60 °C)



Magnetite Fe_3O_4



cubic inverse spinel structure

$$a = 8.394 \text{ \AA}$$

$$T_C \sim 847 \text{ K}$$

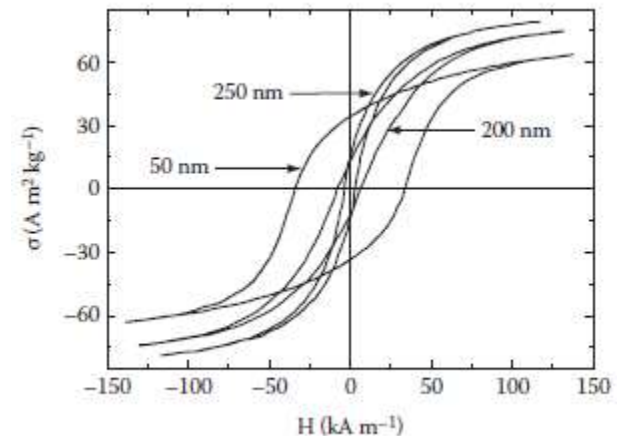
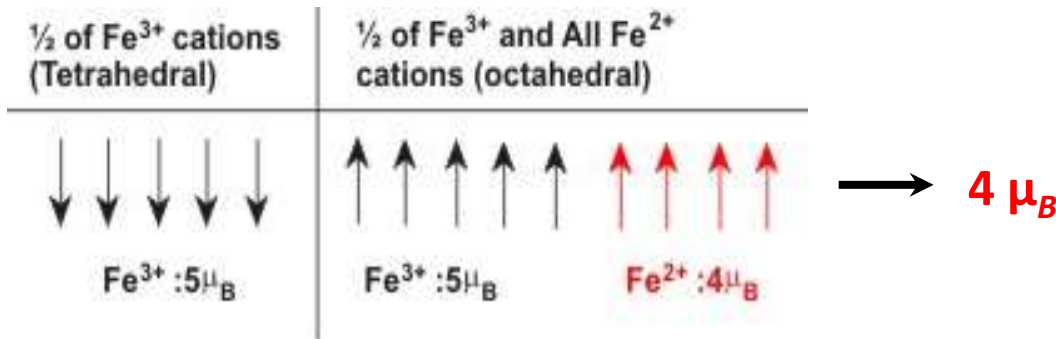
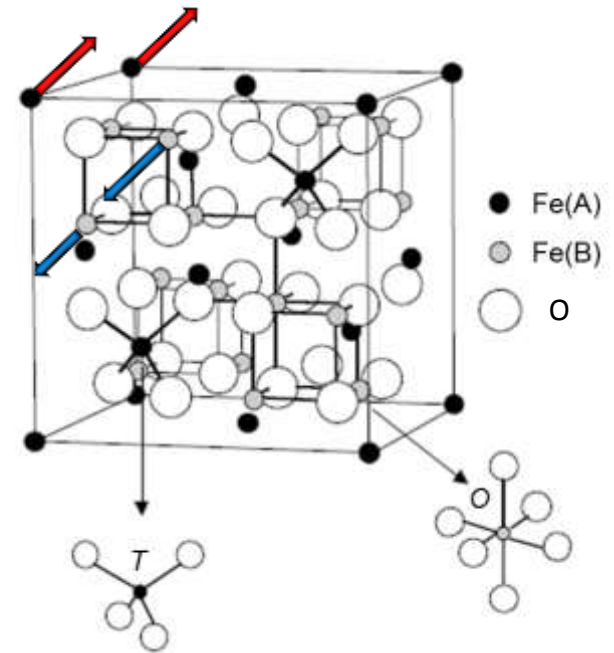
monoclinic structure below $T_V \sim 125 \text{ K}$

$$a = 5.912 \text{ \AA}$$

$$b = 5.945 \text{ \AA}$$

$$c = 8.388 \text{ \AA}$$

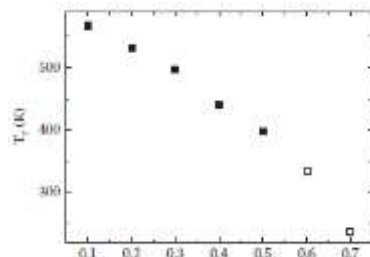
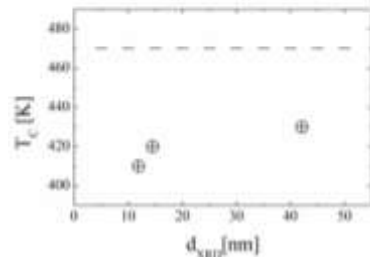
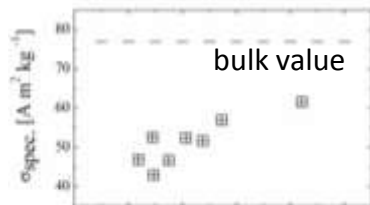
- 8 formulas in elementary cell
- 8 tetrahedral A sites, 16 octahedral B sites
- ferrimagnetic ordering
- phase structural transition and spin reorientation at T_V $\langle 111 \rangle \rightarrow \langle 100 \rangle$



Substituted $M_xFe_{3-x}O_4$ -based nanoparticles

Parameters' dependence:

- substituent and its properties
- distribution over the crystallographic sites
- particle size



x

$T_c(Fe_3O_4 \text{ bulk})$ 847 K $4 \mu_B$
 $T_c(CoFe_2O_4 \text{ bulk})$ 769 K $3 \mu_B$

Bulk... what about nano?

$Zn^{2+} \rightarrow$ A sites

$Co^{2+} \rightarrow$ B sites

MS@RT:

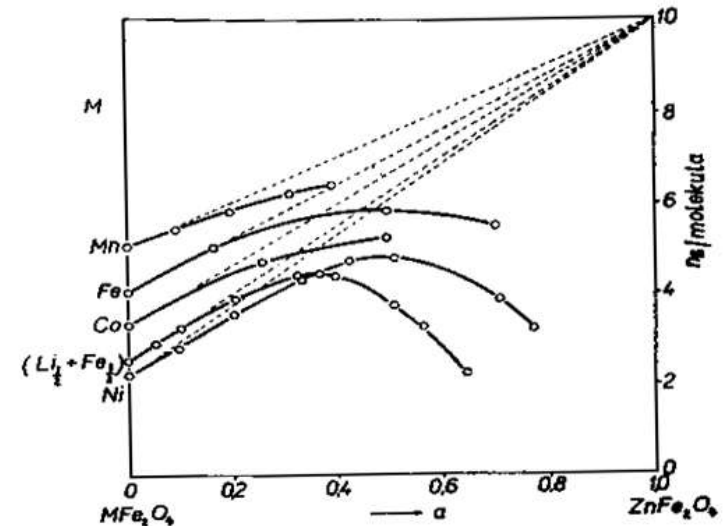
1 sextet for Fe in A sites
 4 sextets for Fe in B sites

MS@LHT:

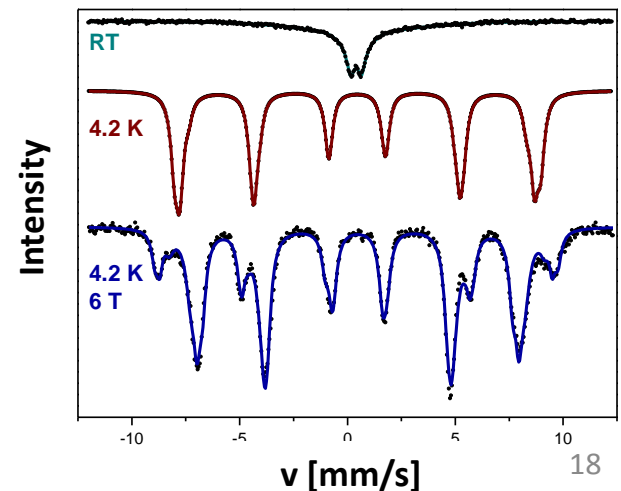
splitting
 IS ... valency

B_{hf} ... cation distribution

$M_{1-x}Zn_xFe_2O_4$ (bulk)



$Co_{1-x}Zn_xFe_2O_4/SiO_2$ (10 nm)



To sum up...

...perspectives.

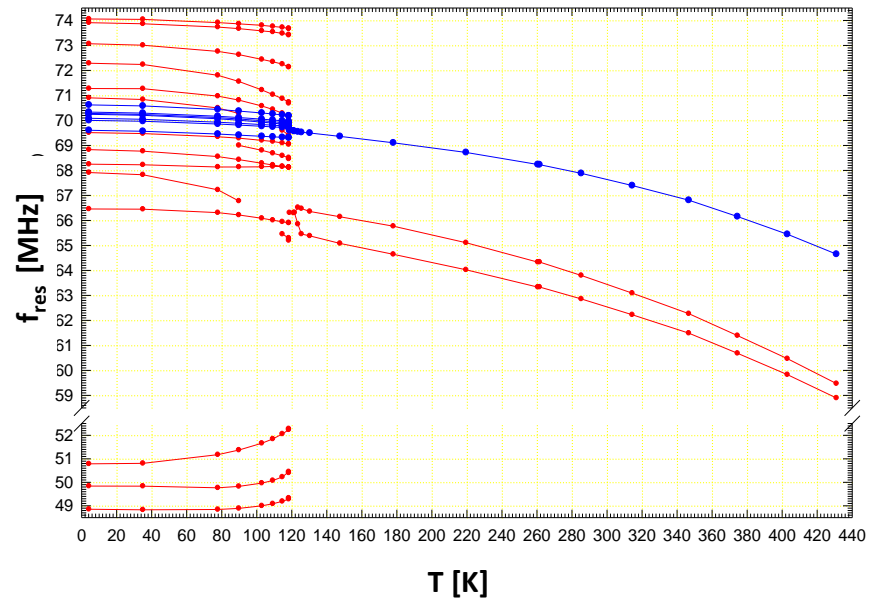
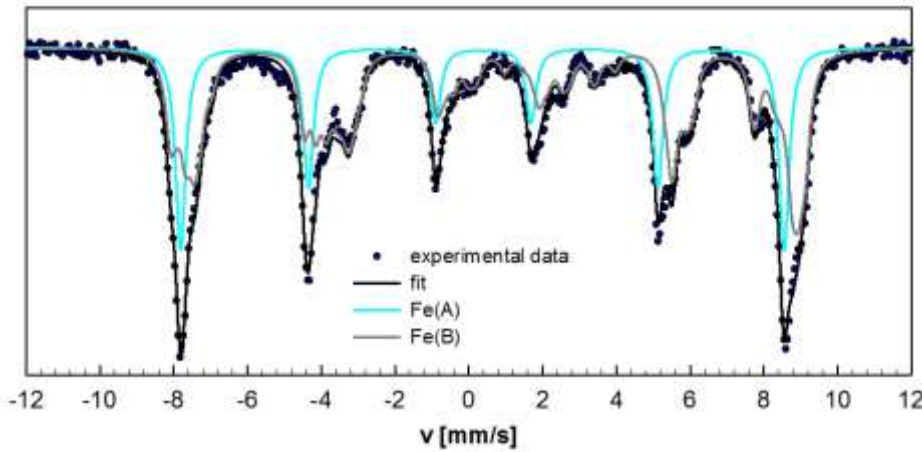
- 1) Optimal method to synthesize the pure ϵ -Fe₂O₃ nanoparticles
- 2) First measurements at ϵ -Fe₂O₃ and Co_{1-x}Zn_xFe₂O₄ nanoparticles
- 3) Substitutions of Fe in ϵ -M_xFe_{2-x}O₃ and spinel M_xFe_{3-x}O₄ nanoparticles by diamagnetic ions to increase net magnetization and relaxivity parameter for MR imaging
- 4) Substitutions of Fe by diamagnetic ions of two species into the same material

Thank you for your attention!

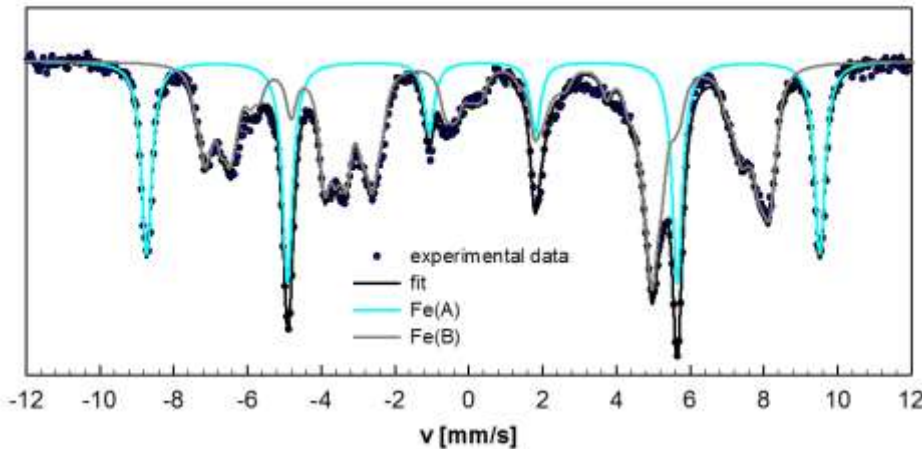
...waiting for your questions/comments.

Mössbauer study of magnetite Fe_3O_4

Pure powder sample of Fe_3O_4 ($T=4,2\text{ K}$, $B=0\text{ T}$)

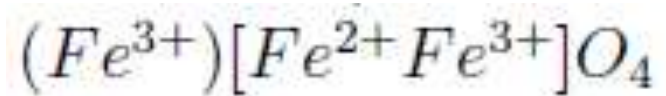


Pure powder sample of Fe_3O_4 ($T=4,2\text{ K}$, $B=6\text{ T}$)



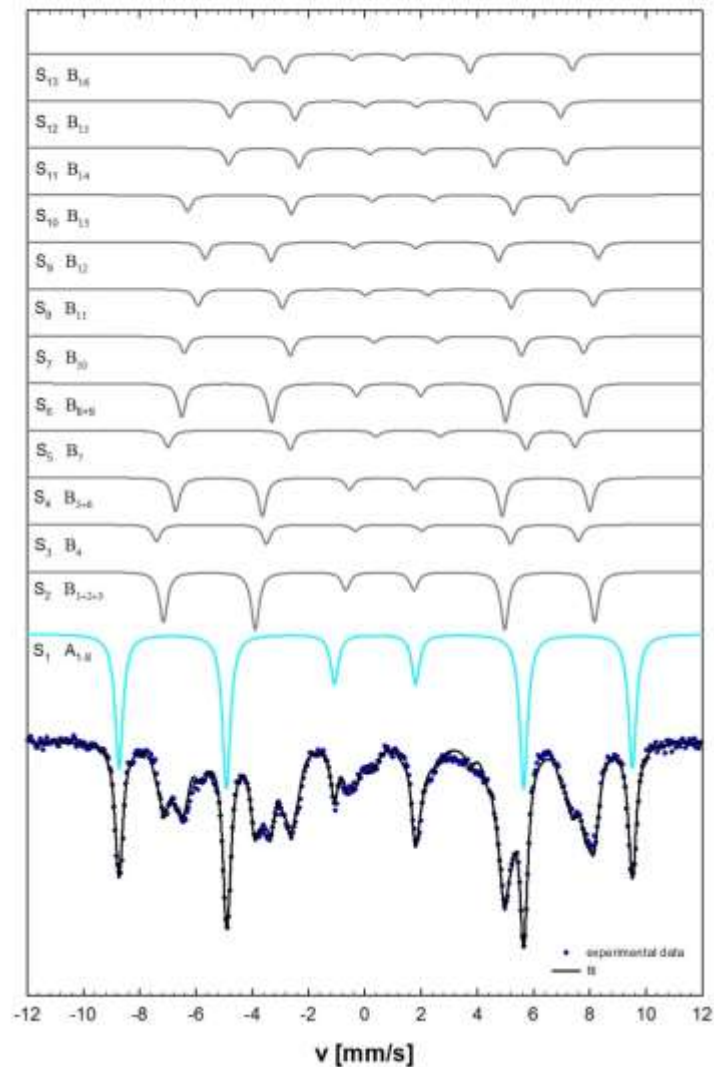
A

B



Mössbauer study of magnetite Fe_3O_4

Powder sample of Fe_3O_4 of 99,99 % purity
($T=4,2$ K, $B=6$ T)

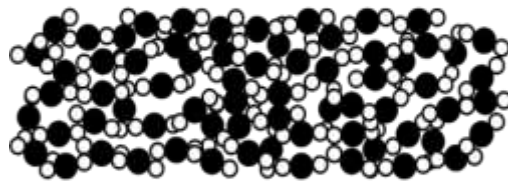
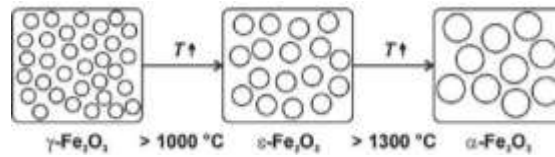
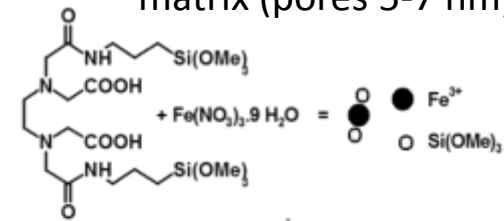


| komponenta | Fe pozícia | B = 0 T | | | B = 6 T | | intenzita [y:1] |
|------------|--------------------|--------------|----------------------|-----------------|----------------------|------------------|--------------------|
| | | IS [mm/s] | Δ_0 [mm/s] | B_{hf} [T] | Δ_0 [mm/s] | B_{eff} [T] | |
| S_1 | A ₁₋₈ | 0,38 | -0,02 | 50,8 | 0,02 | 56,7 | 8 |
| S_2 | B ₁₊₂₊₃ | 0,52 | 0,00 | 53,5 | -0,03 | 47,6 | 3 |
| S_3 | B ₄ | 0,47 | 0,12 | 52,4 | -0,75 | 46,6 | 1 |
| S_4 | B ₅₊₆ | 0,63 | -0,06 | 51,6 | 0,02 | 45,7 | 2 |
| S_5 | B ₇ | 0,89 | -0,71 | 50,8 | -1,30 | 45,0 | 1 |
| S_6 | B ₈₊₉ | 0,76 | -0,22 | 50,5 | -0,18 | 44,6 | 2 |
| S_7 | B ₁₀ | 1,07 | -0,69 | 49,9 | -0,78 | 44,1 | 1 |
| S_8 | B ₁₁ | 1,12 | -0,25 | 49,5 | -0,04 | 43,7 | 1 |
| S_9 | B ₁₂ | 1,02 | -0,99 | 49,3 | 0,60 | 43,4 | 1 |
| S_{10} | B ₁₃ | 0,93 | -0,86 | 48,2 | -0,83 | 42,4 | 1 |
| S_{11} | B ₁₄ | 1,15 | 1,24 | 36,8 | 0,03 | 37,3 | 1 |
| S_{12} | B ₁₅ | 1,00 | 1,83 | 36,2 | 0,16 | 36,5 | 1 |
| S_{13} | B ₁₆ | 1,07 | 2,06 | 35,4 | 1,25 | 35,3 | 1 |

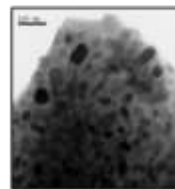
Nanoparticles of ϵ -phase Fe_2O_3

Methods of preparation:

- hydrothermal method
- microemulsion method
- sol-gel synthesis followed by heat treatment (HT)
- impregnation of mesoporous amorphous silica matrix (pores 5-7 nm) by nitrates $\text{Fe}(\text{NO}_3)_3$ + HT



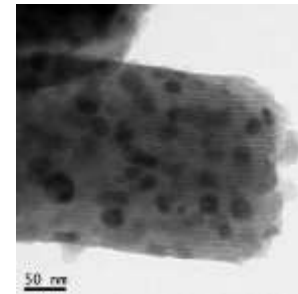
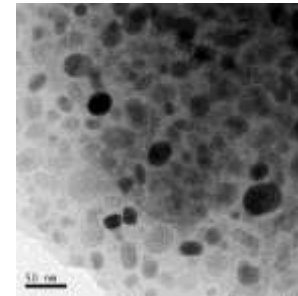
annealing



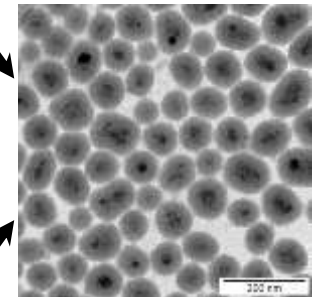
powder sample

$\text{Fe}_2\text{O}_3/\text{SiO}_2$

sol-gel



impregnation



$\epsilon\text{-Fe}_2\text{O}_3$ coated in SiO_2

HRTEM of $\epsilon\text{-Fe}_2\text{O}_3$ nanoparticle

